

NEW PRASEODYMIUM(III) AND *d*-ELECTRON METALS TUNGSTATES OF THE FORMULA MPr₂W₂O₁₀ (M=Mn, Co, Cd)

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Three new compounds MPr₂W₂O₁₀ (M=Mn, Co, Cd) were prepared by the solid-state reaction. They are isostructural and crystallize in the orthorhombic system. MPr₂W₂O₁₀ (M=Mn or Co) melt incongruently above 1150°C and the solid product of melting is Pr₂W₂O₉. The CdPr₂W₂O₉ compound starts decomposing in the solid-state at 1156°C to Pr₂W₂O₉ and CdO.

Keywords: DTA-TG, *d*-electron metal tungstates, IR, praseodymium(III) tungstate, XRD

Introduction

Rare earth ions are characterised by incompletely filled the 4*f* orbitals. They can absorb an excitation energy to be at the excited state and then return to the ground state resulting in an emitting a radiation in the visible region. The *f-f* transitions in RE³⁺ ions have been found practical applications in laser host materials. As important optical materials, tungstates and molybdates of rare-earth metals have been used in quantum electronics [1], scintillators in medical devices and as phosphors (fluorescent lamps, X-ray detectors, cathode ray tubes) [2–5]. Inorganic pigments based on rare-earth compounds ((Bi₂O₃)_{1-x}(Er₂O₃)_x [6], (Bi₂O₃)_{1-x}(Ho₂O₃)_x [7]) are characterized by high covering and colouring power. They are materials, which are friendly for an environment and are often used in a production of paints, printing inks and plastics [6, 7].

In the present study, the new praseodymium(III) and *d*-electron metals tungstates with the formula MPr₂W₂O₁₀ (M=Mn, Co, Cd) have been prepared by the high-temperature solid-state reaction. This method of synthesis is often used to prepare phosphors, catalysts and other solid materials with interesting properties [8–10]. Thermal and some spectroscopic properties of MPr₂W₂O₁₀ have been investigated.

Experimental

Praseodymium(III) tungstate (Pr₂WO₆) and *d*-electron metal tungstates MWO₄ (M=Mn, Co and Cd) were used as the starting materials. Pr₂WO₆ was prepared using the solid-state reaction technique at high temperatures between Pr₆O₁₁ (99.9%, Aldrich) and WO₃ (99.9%, Fluka). Divalent metal tungstates were

obtained by a calcination in air the following mixtures: WO₃+Mn(NO₃)₂·4H₂O (99.9%, POCh), WO₃+CoSO₄·7H₂O (99.9%, Aldrich) and WO₃+3CdSO₄·8H₂O (99.99%, POCh). The mixtures of Pr₂WO₆ with MWO₄ were prepared with the following praseodymium(III) tungstate contents: 20.00; 25.00; 33.33; 45.00; 50.00; 55.00; 66.67; 75.00 and 90.00 mol%. The mixtures of Pr₂WO₆ with MWO₄ (M=Mn, Co) were heated in air, in 12 h periods at the following temperatures: 900, 950, 1000, 1025, 1050, 1075 and 1080°C. The CdWO₄+Pr₂WO₆ mixtures were calcinated in the following heating stages: 900°C (12 h); 950°C (12 h); 1000°C (12 h); 1025°C (12 h) and 1050°C (3–12 h). After each heating period the samples were gradually cooled to ambient temperature, weighed and grinded. No mass changes of samples were observed after each heating stage. A reaction progress was controlled on the base of the results of XRD analysis for the heated samples. After the final heating stage the samples were examined by DTA-TG and IR methods.

X-ray powder diffraction patterns of the analyzed samples were recorded on DRON-3 diffractometer using CuK_α radiation ($\lambda=0.15418\text{ nm}$). The scans were performed in the 2Θ range from 10 to 45° (step 0.02° and 1 s step⁻¹).

DTA-TG examinations were recorded with a Mettler Toledo TGA/SDTA 851 apparatus. These measurements were carried out within the temperature range 25–1400°C, in nitrogen (gas flow 15 mL min⁻¹), using corundum crucibles and at the heating rate of 10 K min⁻¹.

The IR spectra were recorded on a Specord M-80 spectrometer. The samples were pressed in pellets with KBr in the mass ratio of 1:100.

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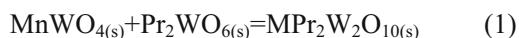
Results and discussion

Reactivity in the solid-state between Pr₂WO₆ and MWO₄

The results of XRD analysis for the samples obtained after the final heating stage showed that the initial components are not mutually inert in air. Only one set of diffraction lines was identified on each powder diffraction pattern of samples obtained after heating equimolar MWO₄+Pr₂WO₆ mixtures. This set of reflexions cannot be considered to be characteristic for any of the used starting materials or other known praseodymium(III) tungstates (i.e., Pr₂W₂O₉, Pr₆W₂O₁₅, Pr₈W₅O₂₇, Pr₆WO₁₂ and Pr₂W₃O₁₂ [11–18]) and neither for any known praseodymium oxides which form a homologous series with the general formula Pr_nO_{2n-2} ($n=4, 5-6, 7, 8, 9, 10, 12, \infty$) [19–25] nor any known M_xO_y ($M=Mn, Co, Cd$) [26–30]. XRD measurements of the samples, obtained by heating the mixtures comprising initially up to 50.00 mol% of Pr₂WO₆ showed the presence of two

solid phases, i.e. MWO₄ and the new phases the diffraction lines sets of which were observed in diffractions patterns of samples comprising initially 50.00 mol% of Pr₂WO₆ and 50.00 mol% of MWO₄. In the concentration range above 50.00 mol% of Pr₂WO₆ it was found that two solid phases, i.e. Pr₂WO₆ and the new phases synthesized by heating equimolar mixtures Pr₂WO₆ with MWO₄, occurred in the samples on treatment.

On the base of the above information it can be stated that Pr₂WO₆ reacted with MWO₄ ($M=Mn, Co$ and Cd) to the MPr₂W₂O₁₀ compounds. The reaction of Pr₂WO₆ with MWO₄ can be described by the following equation:



Crystallography (from XRD data)

The powder diffraction patterns of samples obtained by calcination equimolar mixtures of MWO₄ with Pr₂WO₆ were subjected to indexing of the MPr₂W₂O₁₀ com-

Table 1 Results of indexing MPr₂W₂O₁₀ ($M=Mn, Co, Cd$) powder diffraction patterns

No.	MnPr ₂ W ₂ O ₁₀			CoPr ₂ W ₂ O ₁₀			CdPr ₂ W ₂ O ₁₀			<i>h</i>	<i>k</i>	<i>l</i>
	<i>d</i> _{obs} /nm	<i>d</i> _{cal} /nm	100 <i>I</i> / <i>I</i> ₀	<i>d</i> _{obs} /nm	<i>d</i> _{cal} /nm	100 <i>I</i> / <i>I</i> ₀	<i>d</i> _{obs} /nm	<i>d</i> _{cal} /nm	100 <i>I</i> / <i>I</i> ₀			
1	0.84090	0.84098	3	0.84011	0.84086	4	0.84090	0.84102	5	0	0	1
2	0.61636	0.61684	3	0.61551	0.61635	3	0.62109	0.62053	2	1	1	0
3	0.48531	0.48600	4	0.48531	0.48567	3	0.48610	0.48622	2	0	3	0
4	0.42827	0.42822	5	0.42786	0.42797	5	0.42950	0.42951	4	1	2	1
5	0.42045	0.42049	23	0.42045	0.42043	24	0.42065	0.42051	18	0	0	2
6	0.34743	0.34744	9	0.34729	0.34732	10	0.34823	0.34811	10	1	1	2
7	0.33143	0.33147	6	0.33119	0.33120	6	0.33362	0.33374	5	2	1	0
8	0.31788	0.31799	100	0.31788	0.31787	100	0.31821	0.31806	100	0	3	2
9	0.30841	0.30842	11	0.30820	0.30818	11	0.31020	0.31027	18	2	2	0
10	0.30010	0.30017	8	0.29991	0.29998	7	0.30069	0.30069	9	1	4	1
11	0.28956	0.28956	79	0.28938	0.28936	77	0.29113	0.29109	72	2	2	1
12	0.28036	0.28033	8	0.28036	0.28029	8	0.28053	0.28034	5	0	0	3
13	0.27607	0.27528	2				0.27533	0.27530	1	0	1	3
14	0.26788	0.26805	3	0.26788	0.26786	2	0.26843	0.26845	2	1	5	0
15	0.26040	0.26031	28	0.26018	0.26017	28	0.26151	0.26142	29	2	1	2
16	0.24873	0.24878	7	0.24853	0.24859	6	0.24980	0.24978	3	2	4	0
17	0.24410	0.24423	2	0.24410	0.24416	4	0.24423	0.24448	1	1	2	3
18	0.24289	0.24300	17	0.24276	0.24283	17	0.24314	0.24311	16	0	6	0
19	0.23854	0.23856	6	0.23842	0.23839	6	0.23946	0.23945	8	2	4	1
20	0.23336	0.23345	2	0.23330	0.23330	3				0	6	1
21	0.22605	0.22603	2	0.22594	0.22591	2				1	5	2
22	0.22147	0.22145	5	0.22132	0.22129	4	0.22226	0.22218	3	2	5	0
23	0.21419	0.21411	5	0.21400	0.21398	5	0.21482	0.21475	4	2	4	2
24	0.21132	0.21124	7	0.21118	0.21116	6	0.21165	0.21143	5	1	4	3
25	0.20754	0.20744	2	0.20736	0.20735	2	0.20818	0.20801	2	2	2	3
26	0.20570	0.20561	3	0.20548	0.20545	2	0.20686	0.20685	2	3	3	0

pounds. First successive diffraction lines recorded within 2Θ (CuK_α) $10\text{--}45^\circ$ region were selected for indexing procedure performed by means of the POWDER program [31, 32]. Results of indexing the diffraction patterns of $\text{MPr}_2\text{W}_2\text{O}_{10}$ have been presented in Table 1. Table 2 shows the values of parameters of the unit cells as well as the values of experimental (obtained by degassing the samples and hydrostatic weighing in a pycnometric liquid – CCl_4) and calculated density for the new phases. The obtained phases are isostructural and crystallize in the orthorhombic system. On the base of a comparison of the powder diffraction patterns the author suggest that $\text{MPr}_2\text{W}_2\text{O}_{10}$ are not isostructural with the compounds characterized by an identical type of a chemical formula (i.e. $\text{CoRE}_2\text{W}_2\text{O}_{10}$ where $\text{RE}=\text{Y}$, Dy , Ho , Er [33] and $\text{CuRE}_2\text{W}_2\text{O}_{10}$ where $\text{RE}=\text{Nd}$, Sm , Eu , Gd [34, 35]).

Thermal properties

Figures 1 and 2 show DTA curves of $\text{MPr}_2\text{W}_2\text{O}_{10}$ ($M=\text{Mn}$ or Co). On each DTA curve two endothermic effects were recorded up to 1400°C . No mass losses were recorded on the TG curves (not presented) up to the onsets of the first observed effects on the DTA curves. In order to determining melting behavior of the $\text{MPr}_2\text{W}_2\text{O}_{10}$ ($M=\text{Mn}$, Co) compounds additional experiments have made. Samples of these phases were heated in a furnace at the temperatures higher than the onsets of the first effects, i.e. at 1180°C (Mn); 1170°C (Co). After heating for 4 h the samples were quickly removed from a furnace and quenched to -10°C . On the base of observations of the residues obtained after heating it was found that both phases were melted. The results of XRD analysis made for the samples obtained this way showed that they contained $\text{Pr}_2\text{W}_2\text{O}_9$. Thus, the incongruent melting the $\text{MPr}_2\text{W}_2\text{O}_{10}$ compounds can be described by the following equation:



The endothermic effects with their onsets at 1225°C (Fig. 1, $\text{MnPr}_2\text{W}_2\text{O}_{10}$) and at 1228°C (Fig. 2, $\text{CoPr}_2\text{W}_2\text{O}_{10}$) are associated with melting $\text{Pr}_2\text{W}_2\text{O}_9$ [36–38]. Figure 3 shows DTA-TG curves of

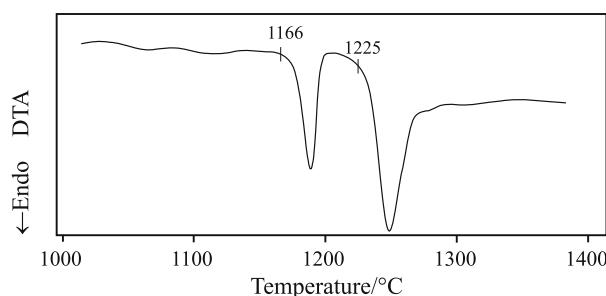


Fig. 1 DTA curve of $\text{MnPr}_2\text{W}_2\text{O}_{10}$

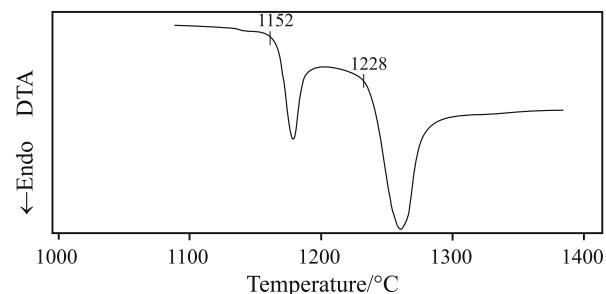


Fig. 2 DTA curve of $\text{CoPr}_2\text{W}_2\text{O}_{10}$

$\text{CdPr}_2\text{W}_2\text{O}_{10}$. Two endothermic effects with their onsets at: 1156 and 1221°C were recorded on the DTA curve of this compound. The first observed effect is accompanied by the mass loss (7.95 mass%, TG curve) starting at the same temperature. Consequently, samples of $\text{CdPr}_2\text{W}_2\text{O}_{10}$ were heated at the temperature higher than the onset of the first effect, i.e. 1180°C and then they were quickly removed from a furnace and next quenched to -10°C . The results of XRD measurements for the samples obtained this way showed that they contained $\text{Pr}_2\text{W}_2\text{O}_9$. It was also ascertained that the ‘freezing’ samples of $\text{CdPr}_2\text{W}_2\text{O}_{10}$ were not melted. Therefore, the decomposition of cadmium and praseodymium(III) tungstate in the solid-state can be described as the following process:



The calculated value of mass loss for the Eq. (3) equals 13.93 mass%. The experimental value is lower than the calculated one. It means that the decomposition process of $\text{CdPr}_2\text{W}_2\text{O}_{10}$ has not been finished under DTA-TG conditions. As in a case of the

Table 2 Calculated parameters of the $\text{MPr}_2\text{W}_2\text{O}_{10}$ unit cells and the values of experimental and calculated density

Compound (colour)	<i>a</i> /nm	<i>b</i> /nm	<i>c</i> /nm	<i>ab</i> ⁻¹	<i>cb</i> ⁻¹	<i>V</i> /nm ³	<i>Z</i>	$\rho_{\text{exp}}/\text{g cm}^{-3}$	$\rho_{\text{cal}}/\text{g cm}^{-3}$
$\text{MnPr}_2\text{W}_2\text{O}_{10}$ (brown)	0.68076(9)	1.4580(1)	0.84097(6)	0.4469	0.5768	0.83472(0)	4	6.82	6.88
$\text{CoPr}_2\text{W}_2\text{O}_{10}$ (green)	0.68021(5)	1.4570(1)	0.84086(3)	0.4669	0.5771	0.83335(6)	4	7.00	6.92
$\text{CdPr}_2\text{W}_2\text{O}_{10}$ (pistachio)	0.68567(2)	1.4586(7)	0.84102(4)	0.4701	0.5766	0.84116(8)	4	7.19	7.28

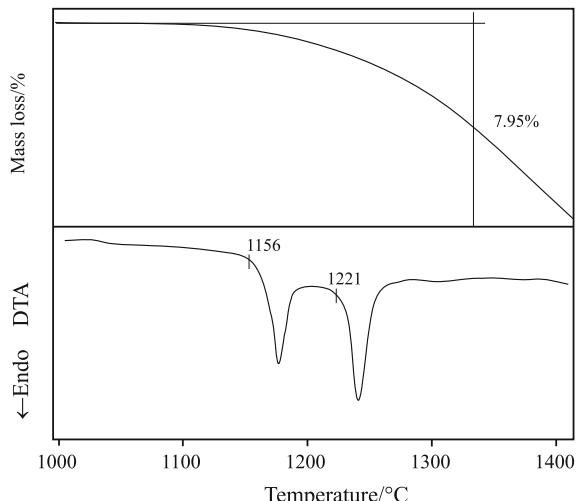


Fig. 3 TG-DTA curves of $\text{CdPr}_2\text{W}_2\text{O}_{10}$

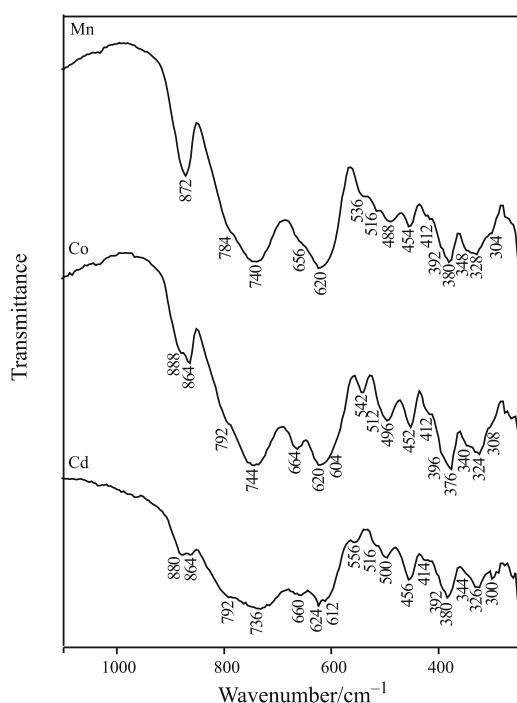


Fig. 4 IR spectra of $\text{MPr}_2\text{W}_2\text{O}_{10}$ compounds

$\text{MPr}_2\text{W}_2\text{O}_{10}$ compounds ($M=\text{Mn}$ or Co), the endothermic effect with its onset at 1221°C is associated with melting $\text{Pr}_2\text{W}_2\text{O}_9$ [36–38].

Infrared spectra

As it is seen from Fig. 4, the spectra of $\text{MPr}_2\text{W}_2\text{O}_{10}$ show big similarity to each other. The absorption bands located at: 872 cm^{-1} (Mn); 884 and 864 cm^{-1} (Co); 880 and 864 cm^{-1} (Cd) could be related to the stretching modes of the W–O bonds in joint WO_6 octahedra by oxygen bridges (three bridges per one octahedron) forming the structural elements

$[(\text{W}_2\text{O}_9)^6]^\infty$ [36, 39]. This structural element was found in structures of the $\text{RE}_2\text{W}_2\text{O}_9$ compounds ($\text{RE}=\text{Pr}$, Nd , Sm – Gd , the crystal structure of these compounds can be described by the following formula $\text{RE}^{[9]}\text{RE}^{[8]}(\text{W}_2^{[6]}\text{O}_9)^\infty$ [36, 39, 40]) and the stretching vibrations of the W–O bonds in the structural elements $[(\text{W}_2\text{O}_9)^6]^\infty$ were observed in IR spectra of these compounds (the region of vibration frequencies 885 – 867 cm^{-1}) [39]. The several absorption bands in the frequencies region 792 – 604 cm^{-1} could be due to the asymmetric stretching vibrations of W–O bonds in joint WO_6 octahedra and also to the oxygen double bridge bonds WOOW [41–44]. According to the literature information [41–44] the absorption bands found in the IR spectra of all analyzed compounds below 556 cm^{-1} could be assigned to the symmetric and also asymmetric deformation modes of W–O bonds in joint WO_6 octahedra as well as to the deformation modes of the oxygen bridges WOOW .

References

- Z. Wang, H. Liang, M. Gong and Q. Su, *J. Alloys Compd.*, 432 (2007) 308.
- J. Wang, F. Liu and H. Zhang, *Mater. Lett.*, 56 (2002) 300.
- M. F. Joubert, A. Remilieux, B. Jacquier, J. Maugnier, B. Boulard, O. Perrot and C. Jacoboni, *J. Non-Cryst. Solids*, 184 (1995) 341.
- Q. Y. Zhang, K. Pita and C. H. Kam, *J. Phys. Chem. Solids*, 64 (2003) 333.
- P. Yang, M. Lü, D. Xü, D. Yuan and G. Zhou, *J. Lumin.*, 93 (2001) 101.
- P. Šulcova and M. Trojan, *J. Therm. Anal. Cal.*, 88 (2007) 111.
- P. Šulcova and M. Trojan, *J. Therm. Anal. Cal.*, 84 (2006) 737.
- A. Blonska-Tabero, *J. Therm. Anal. Cal.*, 88 (2007) 201.
- P. Tabero, *J. Therm. Anal. Cal.*, 88 (2007) 37.
- E. Filipek, *Solidstate Sci.*, 8 (2006) 577.
- S. V. Borisov and R. F. Klevtsova, *Kristallografiya*, 15 (1970) 38 (in Russian).
- M. Yoshimura, H. Morikawa and M. Miyake, *Mater. Res. Bull.*, 10 (1975) 1221.
- K. Nassau, H. J. Levinstein and J. M. Loiacono, *J. Phys. Chem. Solids*, 26 (1965) 1805.
- L. N. Brixner and A. W. Sleight, *Mater. Res. Bull.*, 8 (1973) 1269.
- E. M. Reznik and M. M. Ivanova, *Zh. Neorg. Khim.*, 21 (1976) 522 (in Russian).
- V. K. Trunow and O. V. Kubin, *Zh. Neorg. Khim.*, 22 (1977) 1184 (in Russian).
- G. J. McCarthy, R. D. Fisher, G. G. Johnson and C. E. Gooden, *Nat. Bur. Stand. Spec. Publ.*, 364 (1972) 397.
- I. N. Belaev and L. A. Voronanova, *Zh. Neorg. Khim.*, 19 (1974) 3388 (in Russian).
- Z. C. Kang and L. Eyring, *J. Alloys Compd.*, 275–277 (1988) 721.

- 20 J. Zhang, R. B. Von Dreele and L. Eyring, *J. Solid State Chem.*, 122 (1996) 53.
- 21 J. Zhang, R. B. Von Dreele and L. Eyring, *J. Solid State Chem.*, 118 (1995) 141.
- 22 J. M. Honig, A. F. Clifford and P. A. Faeth, *Inorg. Chem.*, 2 (1963) 791.
- 23 R. B. Von Dreele, L. Eyring, A. L. Bowman and J. L. Yarnell, *Acta Cryst.*, B31 (1975) 971.
- 24 J. Kordis and L. Eyring, *J. Phys. Chem.*, 72 (1968) 2044.
- 25 N. Wakiya, S. Y. Chun, A. Saiki, O. Sakurai, K. Shinozaki and N. Mizutani, *Thermochim. Acta*, 313 (1988) 55.
- 26 A. Cimino and M. Marezio, *J. Phys. Chem. Solids*, 15 (1960) 57.
- 27 M. J. Redman and E. G. Steward, *Nature*, 193 (1962) 867.
- 28 W. L. Roth, *J. Phys. Chem. Solids*, 25 (1964) 1.
- 29 C. A. Barrett and E. B. Evans, *J. Am. Ceram. Soc.*, 47 (1964) 533.
- 30 C. Fong, B. J. Kennedy and M. M. Elcombe, *Z. Kristallogr.*, 209 (1994) 941.
- 31 D. Taupin, *J. Appl. Crystallogr.*, 1 (1968) 87.
- 32 D. Taupin, *J. Appl. Crystallogr.*, 6 (1973) 380.
- 33 E. Tomaszewicz, *J. Therm. Anal. Cal.*, 90 (2007) 255.
- 34 E. Tomaszewicz, A. Worsztynowicz and S. M. Kaczmarek, *Solid State Sci.*, 9 (2007) 43.
- 35 E. Tomaszewicz, J. Typek and S. M. Kaczmarek, to be published.
- 36 A. A. Evdokimov, V. A. Ephremov, V. K. Trunov, I. A. Klejnman and B. F. Dzhyrinskij, *Soedineniya Redkozemelnykh Elementov. Molibdaty, volframaty*. Nauka, Moscow 1991 (in Russian).
- 37 M. Yoshimura and A. Rouanet, *Mater. Res. Bull.*, 11 (1976) 151.
- 38 M. Yoshimura, F. Sibieude, A. Rouanet and M. Foex, *J. Solid State Chem.*, 16 (1976) 219.
- 39 V. I. Tsaryuk and V. F. Zolin, *Spectrochim. Acta*, A57 (2001) 355.
- 40 S. V. Borisov and R. F. Klevtsova, *Kristallografiya*, 15 (1970) 38 (in Russian).
- 41 M. Daturi, G. Busca, M. M. Borel, A. Leclaire and P. Piaggio, *J. Phys. Chem.*, B101 (1997) 4358.
- 42 J. Hanuza, M. Mączka and J. H. van der Maas, *J. Solid State Chem.*, 117 (1995) 177.
- 43 M. Mączka, *J. Solid State Chem.*, 129 (1997) 287.
- 44 J. Hanuza, L. Macalik, M. Mączka, E. T. G. Lutz and J. H. van der Maas, *J. Mol. Struct.*, 511–512 (1999) 2161.

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